
Viscoelastic Material Models

mi@seu.edu.cn

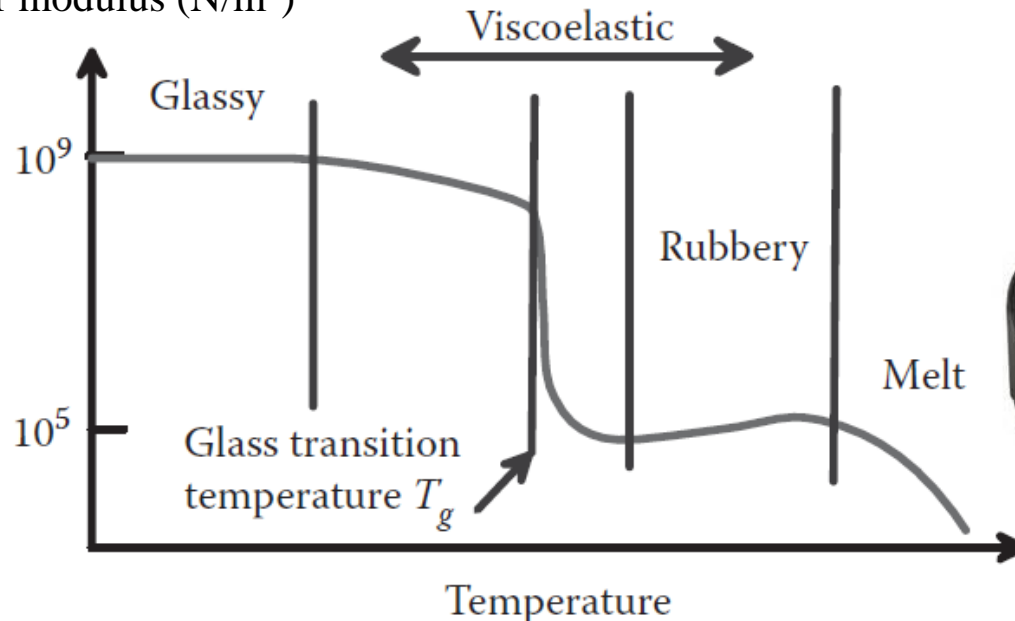
Outline

- Introduction (引言)
- Creep compliance and relaxation modulus (蠕变柔度与松弛刚度)
- Retarded elasticity (滞/粘弹性)
- Steady-state creep (稳态蠕变)
- Elastic vs. viscoelastic materials (弹性与粘弹性材料)
- Maxwell materials (Maxwell模型)
- Kelvin-voigt materials (KV模型)
- Prony series representation (Prony级数模型)
- WLF time/temperature equivalence (WLF时间/温度相当)
- Calibrating viscoelastic models (粘弹性模型校准)
- More sophisticated viscoelastic models (复杂粘弹性模型)
- Generalization to 3D (向三维模型推广)
- Response to harmonic loading (周期载荷的响应)

Introduction

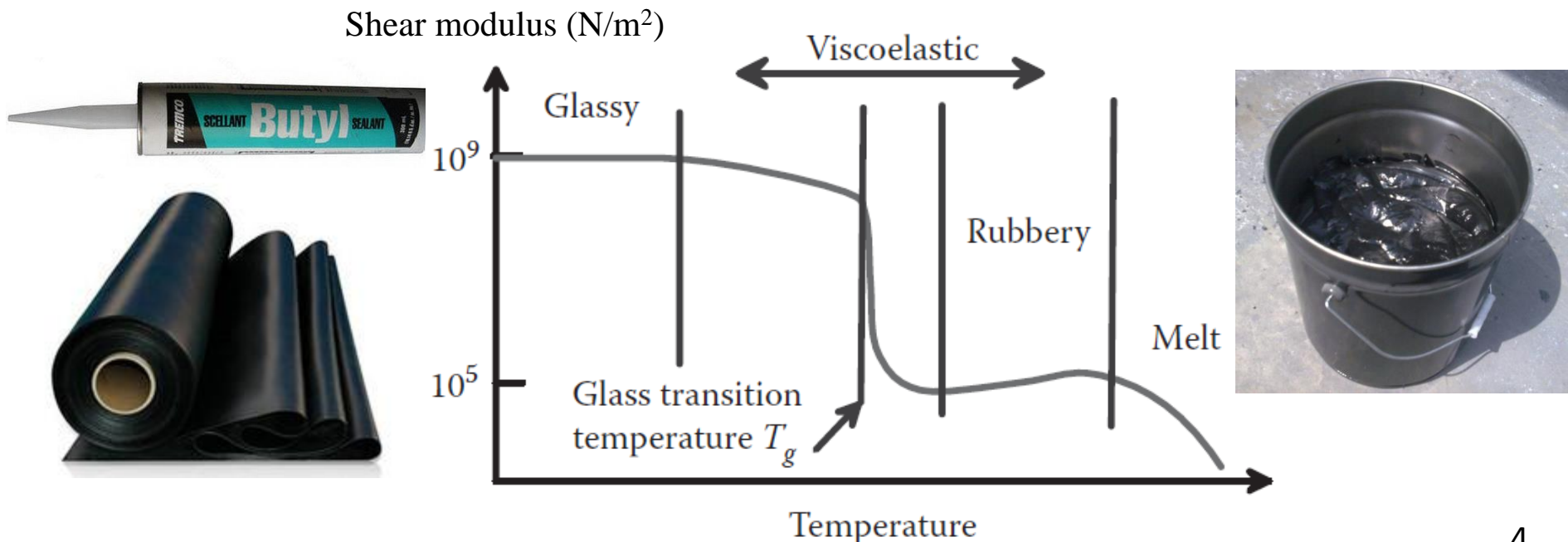
- Amorphous polymers show complex time-dependent behavior when subjected to a history of stress or strain.
- Viscoelasticity theory was developed to approximate this behavior in polymers that are subjected to modest strains (less than 0.5%).
- Polymers strongly resist volume changes at all temperatures. The bulk modulus is comparable to that of metals or covalently bonded solids.
- The shear response of a polymer is strongly temperature dependent.

Shear modulus (N/m^2)



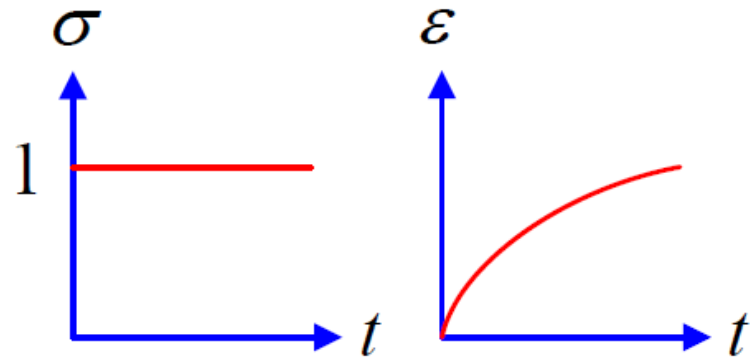
Introduction

- At temperatures near the glass transition, the shear modulus is strongly time dependent. The time dependent shear response can be measured by applying (1) a step load or (2) a harmonic (sinusoidal) load to the specimen.
- The time-dependent modulus of polymers is also temperature dependent. Reducing the temperature is qualitatively equivalent to increasing the strain rate.
- Most amorphous polymers are isotropic.

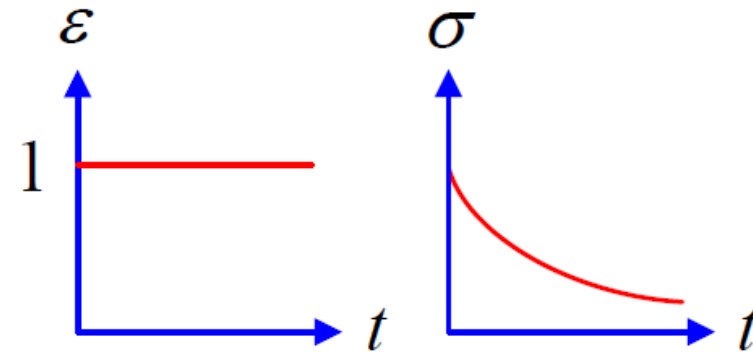


Response to Step Loading

- Creep compliance:
the strain response to
a unit constant stress



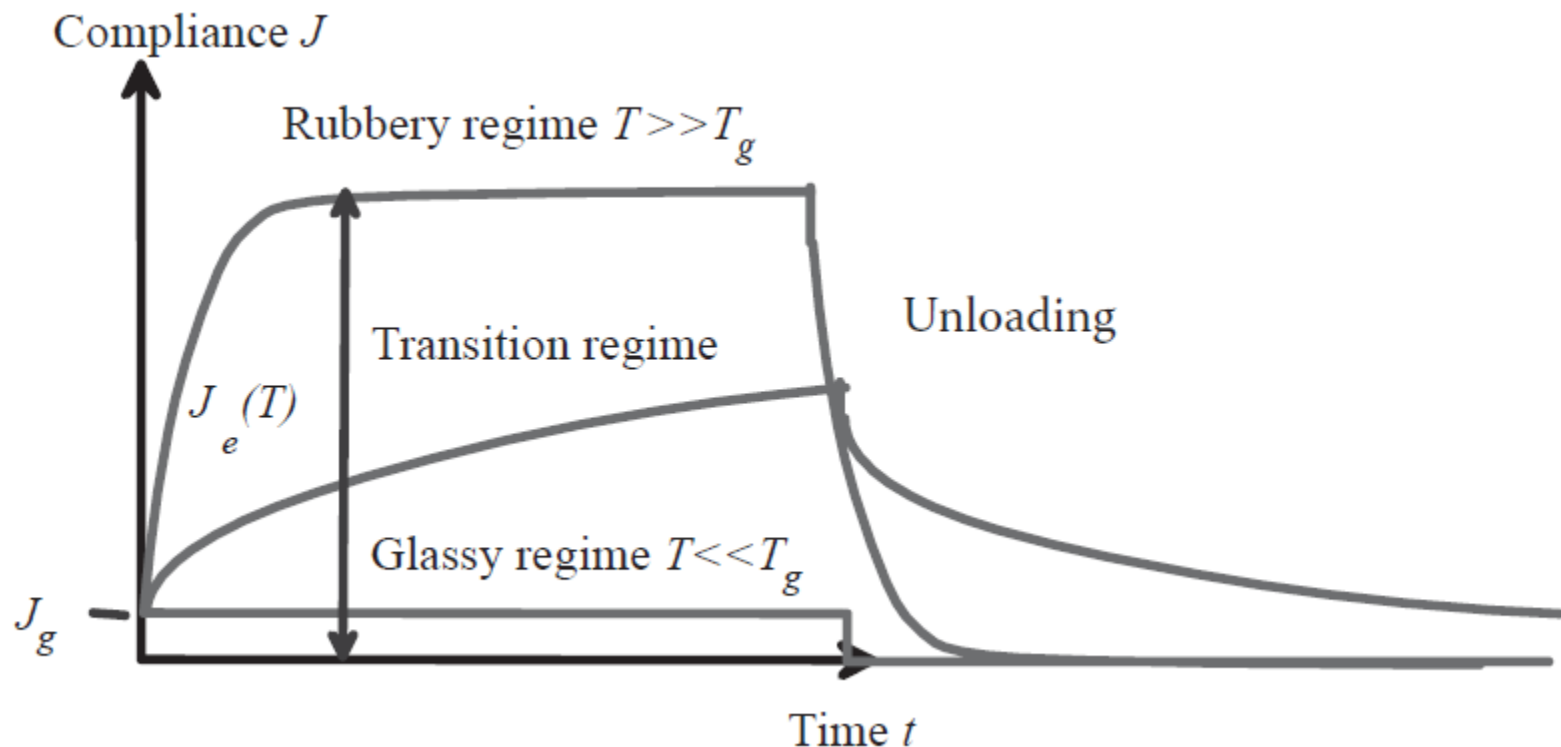
- Relaxation modulus:
the stress response to a
unit constant strain.



- The results of such a test depend on the degree of cross-linking in the polymer.
- Heavily cross-linked materials show “retarded elastic” behavior, whereas un-cross-linked materials show steady-state creep.

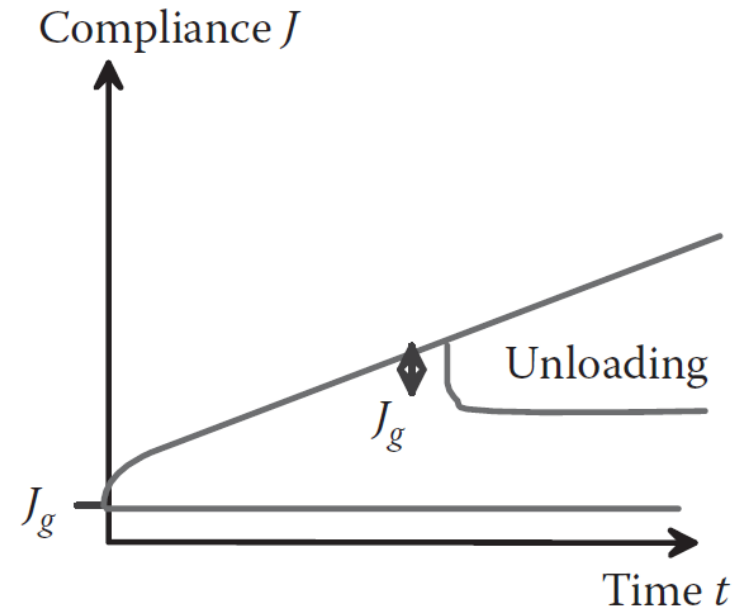
Retarded Elasticity for Crosslinked Polymers

- There is always an instantaneous strain in response to a step change in stress.
- At temperatures significantly below T_g , the solid is essentially elastic.
- At temperatures significantly above T_g , the solid is very compliant.
- For a range of temperatures near T_g , the solid shows a slow transient response.
- The deformation is reversible, although this may take a very long time.



Steady-state Creep for uncrosslinked polymers

- There is always an instantaneous strain in response to a step change in stress, exactly as in crosslinked polymers.
- At temperatures well below T_g , the solid is essentially elastic and has a very low compliance, comparable with J_g .
- At temperatures above T_g , the solid is very compliant. For most practical ranges of loading, the compliance will increase more or less linearly with time. The slope of compliance is strongly temperature dependent.
- Above the glass transition temperature T_g , the deformation is irreversible.



Elastic vs. Viscoelastic Materials

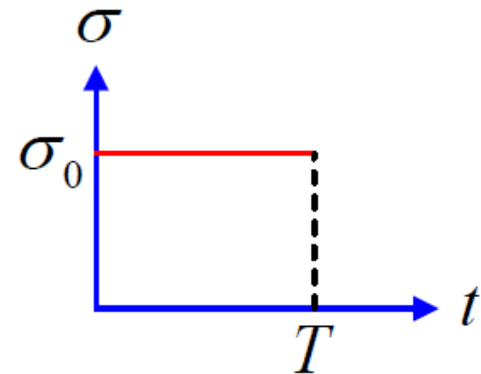
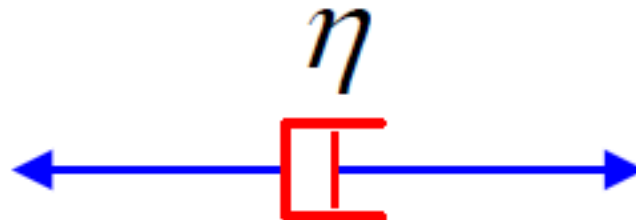
- For a constant loading acting from time 0 to T ,
- The strain responses of elastic materials:

$$\sigma = E\varepsilon \quad \Rightarrow \quad \varepsilon = \frac{\sigma}{E}$$



- The strain responses of **viscoelastic** materials:

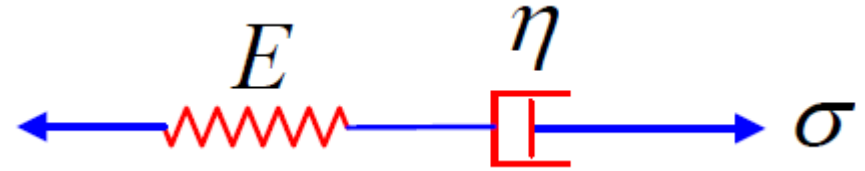
$$\sigma = \eta \dot{\varepsilon} \quad \Rightarrow \quad \dot{\varepsilon} = \frac{\sigma}{\eta}$$



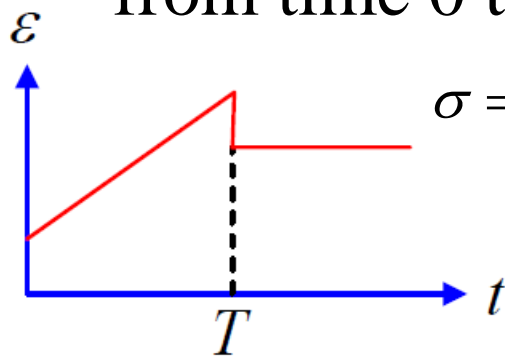
Viscoelastic Models

- **Maxwell model** (steady-state creep of uncrosslinked polymers)

$$\dot{\varepsilon} = \dot{\varepsilon}_E + \dot{\varepsilon}_\eta = \frac{\dot{\sigma}}{E} + \frac{\sigma}{\eta}$$



- Creep compliance due to a constant loading σ_0 acting from time 0 to T



$$\sigma = \sigma_0 \Rightarrow \dot{\sigma} = 0 \Rightarrow \dot{\varepsilon} = \frac{\sigma_0}{\eta} \Rightarrow \varepsilon[t] = \begin{cases} \frac{\sigma_0}{\eta} t + \frac{\sigma_0}{E}, & 0 < t \leq T \\ \frac{\sigma_0}{\eta} T, & t > T \end{cases}$$

$$\Rightarrow J[t] = \frac{\varepsilon[t]}{\sigma_0} = \frac{1}{\eta} t + \frac{1}{E}$$

- Relaxation modulus due to a constant strain ε_0

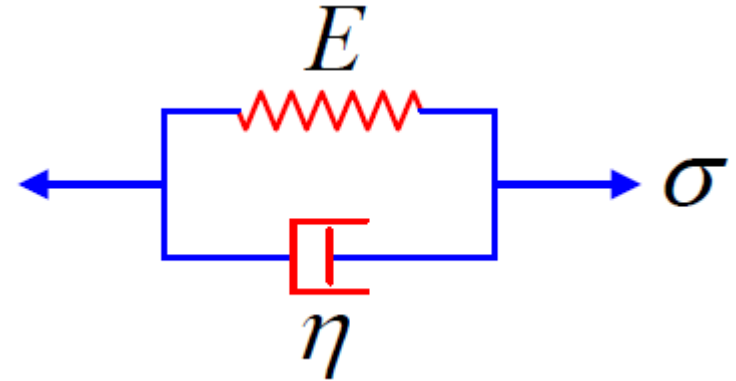
$$\varepsilon = \varepsilon_0 = \text{const} \Rightarrow \frac{\dot{\sigma}}{E} = -\frac{\sigma}{\eta} \Rightarrow \frac{\dot{\sigma}}{\sigma} = -\frac{E}{\eta} \Rightarrow \ln \sigma = -\frac{Et}{\eta} + \ln C \Rightarrow \sigma[t] = C e^{-Et/\eta}$$

$$\sigma[0] = E\varepsilon_0 = C \Rightarrow \sigma[t] = E\varepsilon_0 e^{-Et/\eta} \Rightarrow G[t] = \frac{\sigma[t]}{\varepsilon_0} = E e^{-Et/\eta}$$

Kelvin-Voigt Model (Retarded Elasticity of Crosslinked Polymers)

$$\sigma = \sigma_{\eta} + \sigma_E = \eta \dot{\varepsilon} + E\varepsilon$$

- Creep compliance due to a constant loading σ_0 acting from time 0 to T

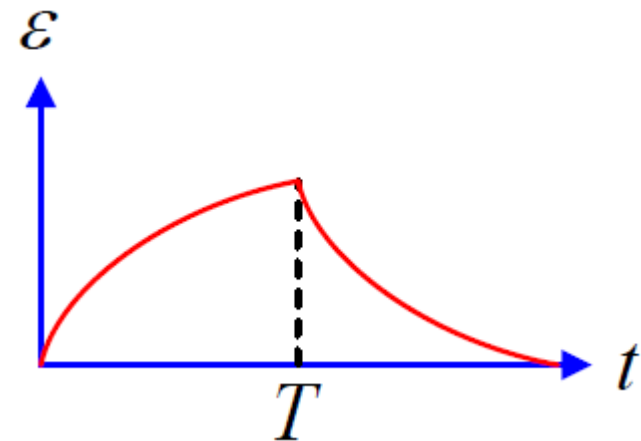


$$0 < t \leq T: \quad \eta \dot{\varepsilon} + E\varepsilon = \sigma_0 \quad \Rightarrow \quad e^{Et/\eta} \left(\dot{\varepsilon} + \frac{E}{\eta} \varepsilon \right) = e^{Et/\eta} \left(\frac{\sigma_0}{\eta} \right)$$

$$\Rightarrow \frac{d}{dt} \left(e^{Et/\eta} \varepsilon \right) = \frac{\sigma_0}{\eta} e^{Et/\eta} \quad \Rightarrow \quad \varepsilon[t] = \frac{\sigma_0}{E} + C e^{-Et/\eta}$$

$$\varepsilon[0] = \frac{\sigma_0}{E} + C = 0 \quad \Rightarrow \quad C = -\frac{\sigma_0}{E} \quad \Rightarrow \quad \varepsilon[t] = \frac{\sigma_0}{E} \left(1 - e^{-Et/\eta} \right)$$

$$t > T: \quad \eta \dot{\varepsilon} + E\varepsilon = 0 \quad \Rightarrow \quad \frac{\dot{\varepsilon}}{\varepsilon} = -\frac{E}{\eta} \quad \Rightarrow \quad \varepsilon[t] = D e^{-Et/\eta}$$



$$\varepsilon[T] = D e^{-ET/\eta} = \frac{\sigma_0}{E} \left(1 - e^{-ET/\eta} \right) \Rightarrow \varepsilon[t] = \frac{\sigma_0}{E} \left(e^{ET/\eta} - 1 \right) e^{-Et/\eta} \Rightarrow \boxed{J[t] = \frac{e^{-Et/\eta}}{E} \left(e^{ET/\eta} - 1 \right)}$$

Standard Linear Solid Model

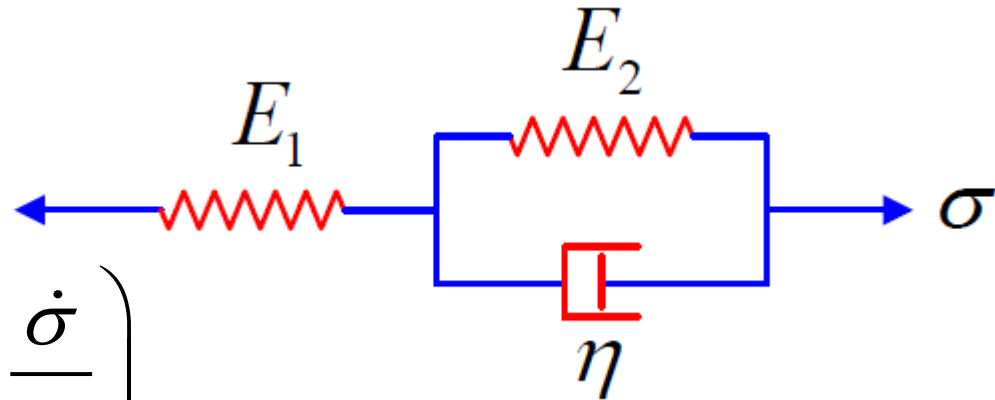
- Spring + Kelvin

$$\sigma = \sigma_{E_1} = \sigma_{E_2} + \sigma_{\eta}$$

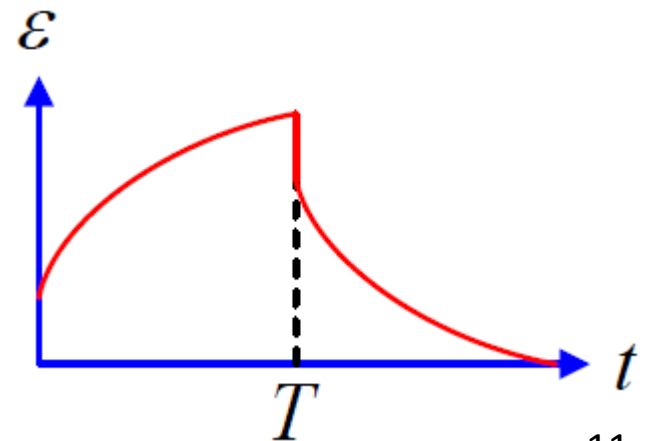
$$= E_2 \varepsilon_2 + \eta \dot{\varepsilon}_2$$

$$= E_2 \left(\varepsilon - \frac{\sigma}{E_1} \right) + \eta \left(\dot{\varepsilon} - \frac{\dot{\sigma}}{E_1} \right)$$

$$\Rightarrow \boxed{(E_1 + E_2) \sigma + \eta \dot{\sigma} = E_1 E_2 \varepsilon + E_1 \eta \dot{\varepsilon}}$$



- Strain response to a constant loading σ_0 acting from time 0 to T

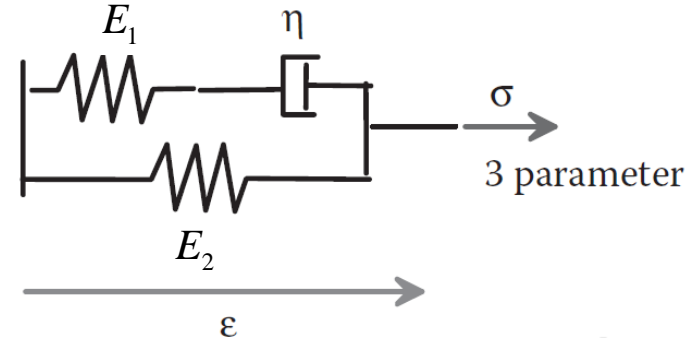


Three-parameter Model

- Spring // Maxwell

$$\dot{\varepsilon} = \dot{\varepsilon}_{E_2} = \dot{\varepsilon}_{E_1} + \dot{\varepsilon}_{\eta} = \frac{\dot{\sigma}_1}{E_1} + \frac{\sigma_1}{\eta} = \frac{\dot{\sigma} - E_2 \dot{\varepsilon}}{E_1} + \frac{\sigma - E_2 \varepsilon}{\eta}$$

$$\Rightarrow \boxed{E_1 \sigma + \eta \dot{\sigma} = E_1 E_2 \varepsilon + (E_1 + E_2) \eta \dot{\varepsilon}}$$



- Relaxation modulus due to a constant strain ε_0

$$\varepsilon = \varepsilon_0 = \text{const} \Rightarrow E_1 \sigma + \eta \dot{\sigma} = E_1 E_2 \varepsilon_0 \Rightarrow e^{E_1 t / \eta} \left(\frac{E_1}{\eta} \sigma + \dot{\sigma} \right) = e^{E_1 t / \eta} \frac{E_1 E_2 \varepsilon_0}{\eta}$$

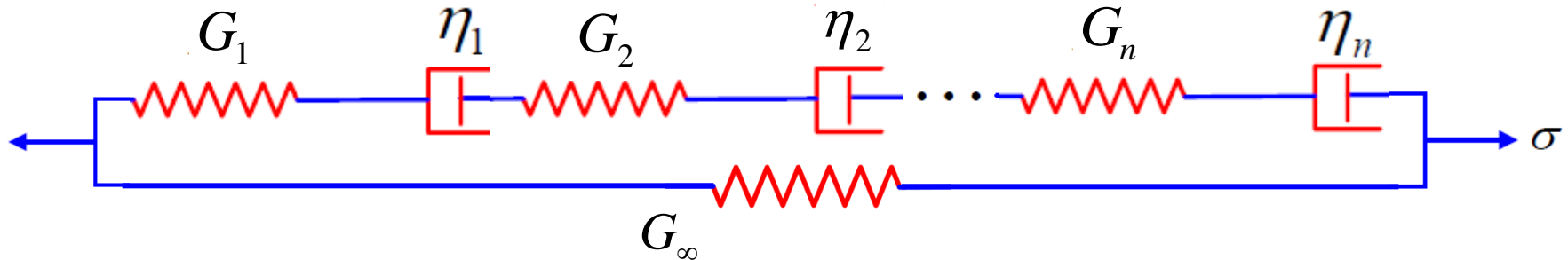
$$\Rightarrow \frac{d}{dt} \left(e^{E_1 t / \eta} \sigma \right) = e^{E_1 t / \eta} \frac{E_1 E_2 \varepsilon_0}{\eta} \Rightarrow e^{E_1 t / \eta} \sigma = e^{E_1 t / \eta} E_2 \varepsilon_0 + C \Rightarrow \sigma[t] = E_2 \varepsilon_0 + C e^{-E_1 t / \eta}$$

$$\sigma[0] = E_2 \varepsilon_0 + C = E_1 \varepsilon_0 + E_2 \varepsilon_0 \Rightarrow C = E_1 \varepsilon_0 \Rightarrow \sigma[t] = E_2 \varepsilon_0 + E_1 \varepsilon_0 e^{-E_1 t / \eta}$$

$$\Rightarrow \boxed{G[t] = \frac{\sigma[t]}{\varepsilon_0} = E_2 + E_1 e^{-E_1 t / \eta}}$$

Prony Series for the Relaxation Modulus

- Spring // lots of Maxwell elements



- Relaxation modulus due to a constant strain ε_0

$$G[t] = \sigma[t]/\varepsilon_0 = G_\infty + \sum_{i=1}^n G_i e^{-t/t_i}, \quad t_i = \eta_i/G_i$$

- These parameters are used directly as the properties of the material.
- The sum of exponentials is known as the *Prony series*.

Williams–Landell–Ferry Time/Temperature Equivalence

- At temperature T_1 , subject a specimen to a step change in shear strain and measure the relaxation modulus.
- Repeat the experiment at several higher temperatures.
- The double log plot demonstrates that, if you simply shift the modulus curves for the higher temperatures to the right, the data collapse onto a single master curve.

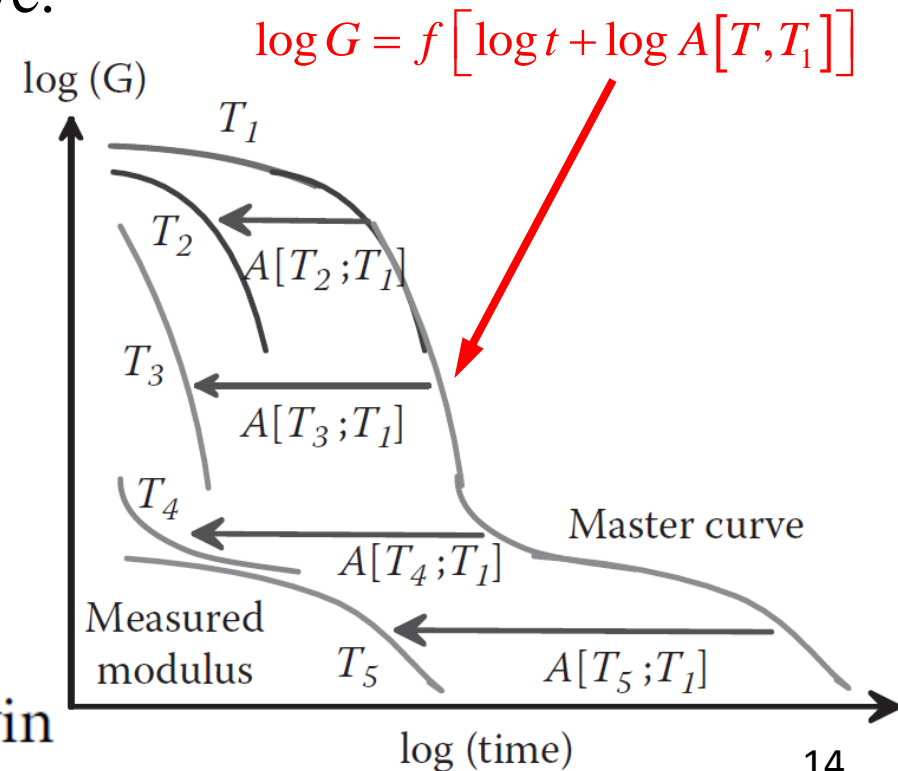
- WLF shift function:

$$\log A[T, T_1] = -\frac{C_1(T - T_1)}{C_2 + T - T_1}$$

- More convenient to use T_g as the reference temperature

$$\log A[T, T_g] = -\frac{C_1^g(T - T_g)}{C_2^g + T - T_g}$$

$$C_1^g \approx 10 - 40 \text{ and } C_2^g \approx 50 - 100 \text{ Kelvin}$$



Calibrating Viscoelastic Models

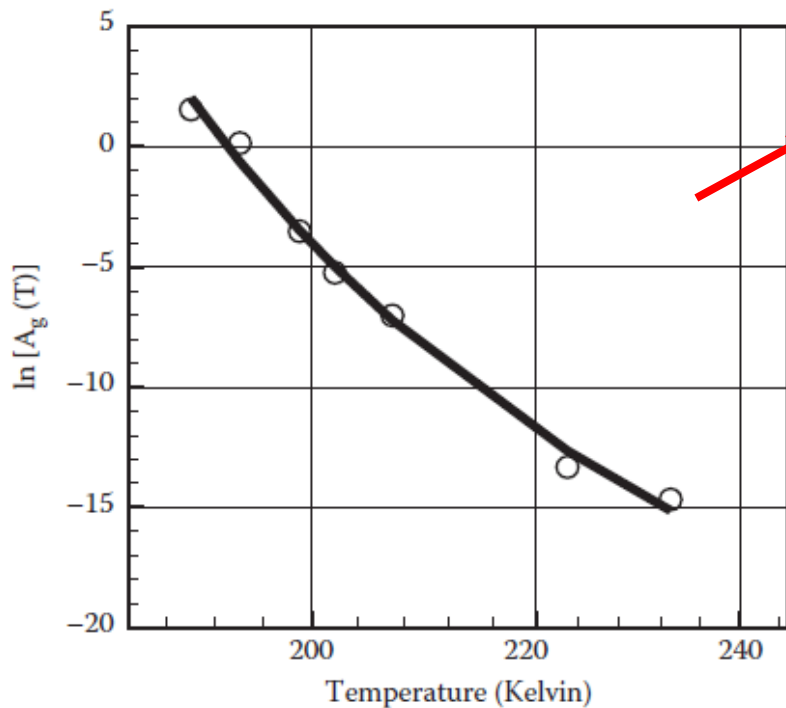


TABLE 3.8 Viscoelastic Properties of Polyisobutylene

WLF parameters: $C_1^g = 36.9$ $C_2^g = 57.6$ Kelvin $T_g = 193$ Kelvin

$G_\infty = 0.143 \text{ MNm}^{-2}$ $G_1 = 7.3 \text{ GNm}^{-2}$ $t_1 = 100 \text{ sec}$

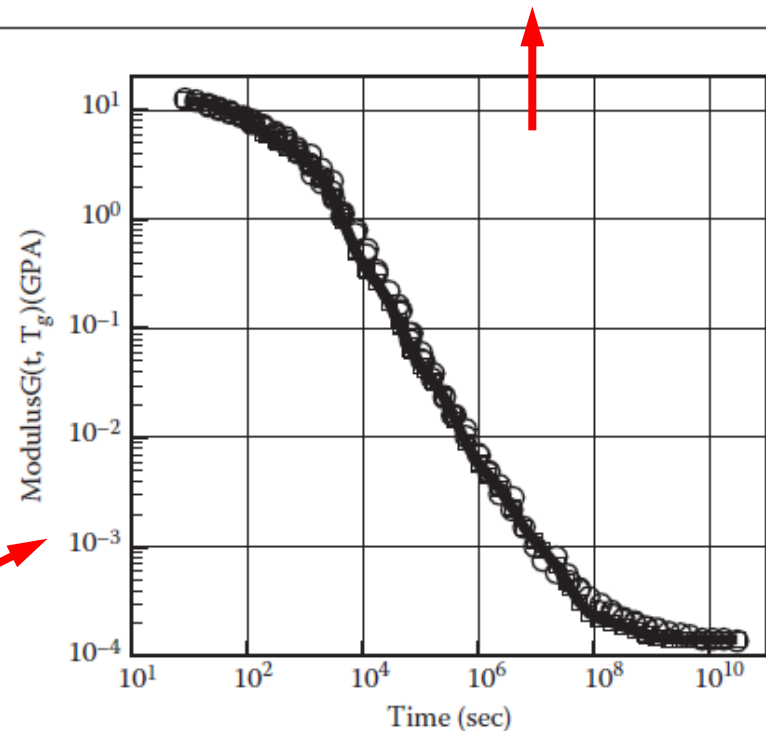
$G_2 = 4.9 \text{ GNm}^{-2}$ $t_2 = 2000 \text{ sec}$ $G_3 = 0.48 \text{ GNm}^{-2}$ $t_3 = 2 \times 10^4 \text{ sec}$

$G_4 = 57 \text{ MNm}^{-2}$ $t_4 = 2 \times 10^5 \text{ sec}$ $G_5 = 6.4 \text{ MNm}^{-2}$ $t_5 = 2 \times 10^6 \text{ sec}$

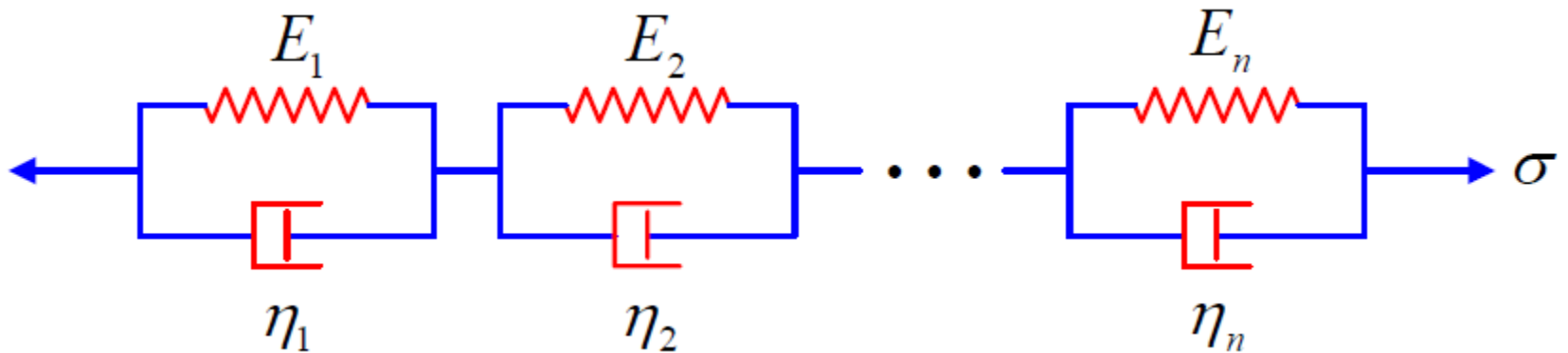
$G_6 = 1.3 \text{ MNm}^{-2}$ $t_6 = 2 \times 10^7 \text{ sec}$ $G_7 = 0.1 \text{ MNm}^{-2}$ $t_7 = 4 \times 10^8 \text{ sec}$

Experimental and fitted WLF shift function for polyisobutylene

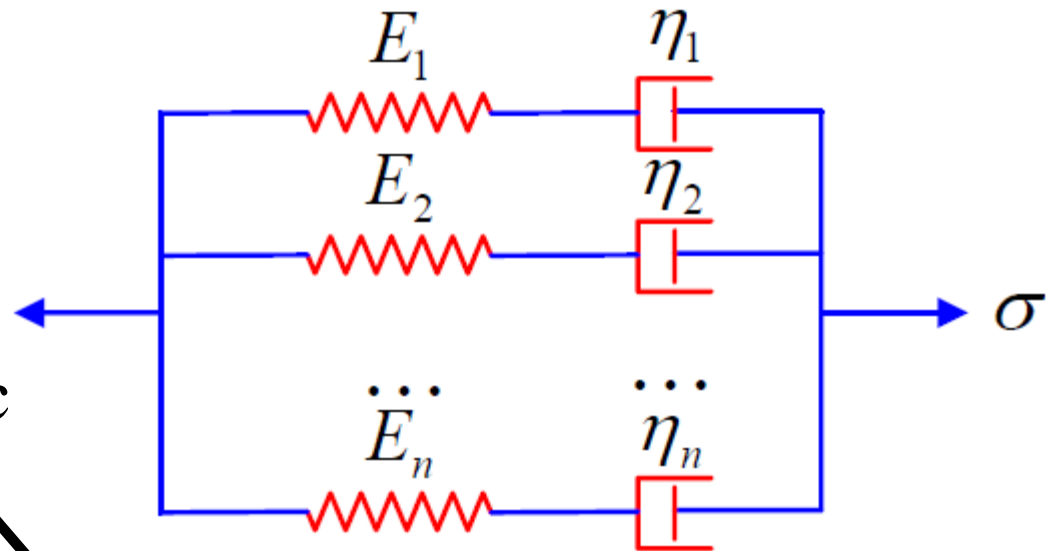
Experimental and fitted relaxation modulus (seven-term Prony series) of polyisobutylene at T_g



More Sophisticated Models



- General mathematical structure of viscoelastic models:



$$\left(p_0 + p_1 \frac{\partial}{\partial t} + p_2 \frac{\partial^2}{\partial t^2} + \dots + p_n \frac{\partial^n}{\partial t^n} \right) \sigma = \left(q_0 + q_1 \frac{\partial}{\partial t} + q_2 \frac{\partial^2}{\partial t^2} + \dots + q_n \frac{\partial^n}{\partial t^n} \right) (E \varepsilon)$$

Generalization to 3D

- 1D linear elastic

$$\sigma = E\varepsilon$$

- 3D linear elastic

$$\sigma_{kk} = 3K\varepsilon_{kk};$$

$$\sigma'_{ij} = 2G\varepsilon'_{ij}.$$

- 1D linear viscoelastic

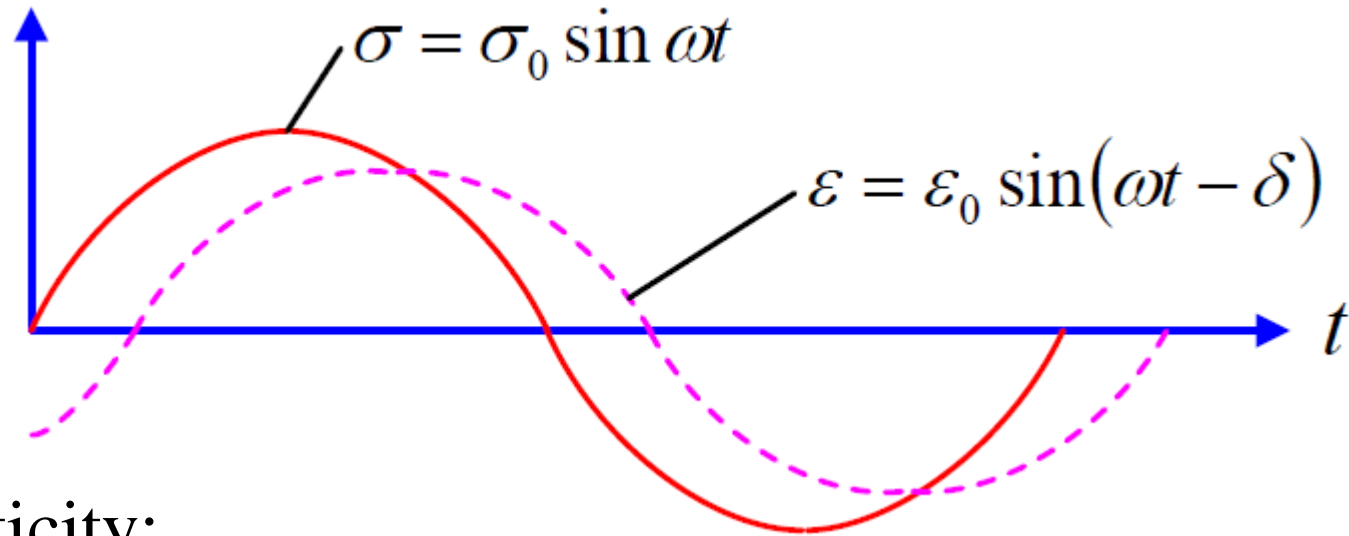
$$P\sigma = Q(E\varepsilon)$$

- 3D linear viscoelastic

$$P\sigma_{kk} = Q(3K\varepsilon_{kk});$$

$$P\sigma'_{ij} = Q(2G\varepsilon'_{ij}).$$

Response to Harmonic Loading



- Elasticity:

$$\sigma = \sigma_0 \sin \omega t, \quad \varepsilon = \frac{\sigma_0}{E} \sin \omega t = \varepsilon_0 \sin \omega t \quad (\text{no delay for strain response})$$

- Viscoelasticity: $\varepsilon = \varepsilon_0 \sin(\omega t - \delta) = \varepsilon_0 \cos \delta \sin \omega t - \varepsilon_0 \sin \delta \cos \omega t$
- Storage compliance: $\varepsilon_0 \cos \delta / \sigma_0$
- Loss compliance: $\varepsilon_0 \sin \delta / \sigma_0$